

## Synthesis and Magnetic Properties of Mesoporous Vanadium Oxide Sulphate

Shigemi Kohiki,\* Hirokazu Shimooka, Syozo Takada, Akihiko Shimizu, Tomohiro Hirakawa, Seiji Takahashi, Hiroyuki Deguchi,<sup>†</sup> and Masaoki Oku<sup>††</sup>

<sup>\*</sup>Department of Materials Science, Kyushu Institute of Technology, Tobata, Kita-kyushu 804-8550

<sup>†</sup>Department of Electrical Engineering, Kyushu Institute of Technology, Tobata, Kita-kyushu 804-8550

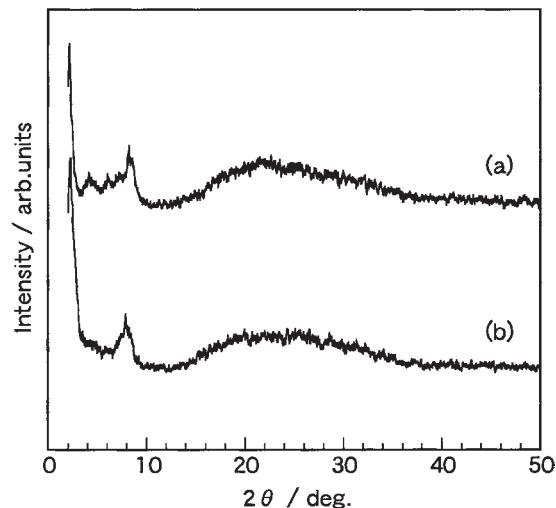
<sup>††</sup>Institute of Materials Research, Tohoku University, Sendai 980-8577

(Received January 28, 2002; CL-020104)

Preparation and magnetic properties of mesostructured  $\text{VOSO}_4$  based materials, which were synthesized using liquid-crystal templating technique, have been investigated. Low temperature magnetic properties of the samples exhibited Langevin paramagnetism with an effective magnetic moment  $\mu_{\text{eff}} = 1.4 \mu_B$  and a paramagnetic Curie point  $\theta = 0 \text{ K}$  without any magnetic transition, which were considerably different from magnetic properties of  $\alpha$ - or  $\beta$ - $\text{VOSO}_4$  crystals.

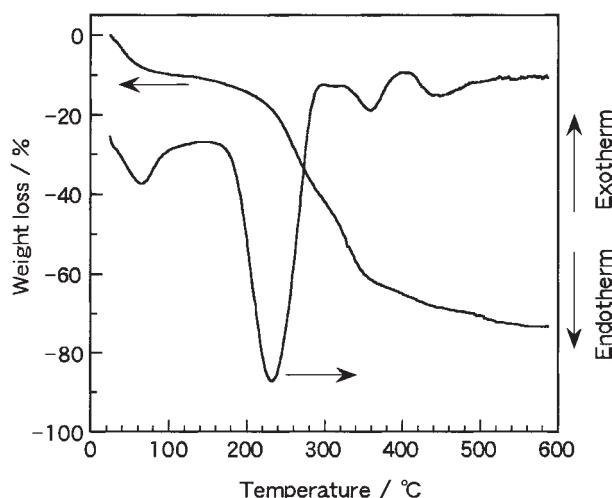
Since reporting of new family of porous materials (MCM-41) with periodic mesopores of 30–40 Å by liquid-crystal templating technique, various mesoporous materials have been synthesized using similar templating method.<sup>1</sup> Whereas mesoporous materials have attracted much attention in a viewpoint of catalyst, we take an interest in those mesoporous materials as templates for quantum dots or wires.<sup>2–4</sup> Vanadium oxide sulphate has two types of polymorphs, i.e.  $\alpha$ - $\text{VOSO}_4$  and  $\beta$ - $\text{VOSO}_4$ , which show different magnetic properties.<sup>5,6</sup>  $\alpha$ - $\text{VOSO}_4$  has a tetragonal structure with chains along  $c$  axis consisted of corner-shared  $\text{VO}_6$  octahedra connected by corner-shared  $\text{SO}_4$  tetrahedra.  $\text{V}-\text{O}$  distance along  $c$  axis is considerably different; thus  $\text{VO}_6$  octahedra regard as  $\text{VO}_5$  pyramid which forms lamellar structure. It is known that the  $\alpha$ -type crystal is ferro- or ferrimagnetic below 4 K and  $\theta = -12 \text{ K}$ .  $\beta$ - $\text{VOSO}_4$  has an orthorhombic structure with zigzag chains of distorted  $\text{VO}_6$  octahedra connected by  $\text{SO}_4$  tetrahedra to form a three-dimensional network.  $\beta$ - $\text{VOSO}_4$  shows antiferromagnetic behaviors with a Néel point  $T_N = 25 \text{ K}$  due to their  $\text{V}-\text{SO}_4-\text{V}$  superexchange interaction. Since both  $\alpha$ - and  $\beta$ - $\text{VOSO}_4$  crystals have features of structural flexibility based on the connection by corner sharing of distorted  $\text{VO}_6$  and  $\text{SO}_4$  units, changes of magnetic properties are expected with changes of the oxidation state and local structure. Thus, if we can prepare mesostructure using crystalline or amorphous  $\text{VOSO}_4$  by means of molecular templating technique, novel magnetic properties associated with decreasing dimension are expected. In this paper, we report synthesis and magnetic properties of mesoporous  $\text{VOSO}_4$  based materials for the first time.

Mesoporous  $\text{VOSO}_4$  based materials were synthesized using  $\text{VOSO}_4 \cdot 3.5\text{H}_2\text{O}$  (Kishida Chemical Co., Ltd.), *n*-cetyltrimethylammonium chloride  $\text{C}_{16}\text{H}_{33}(\text{CH}_3)_3\text{NCl}$  (designated as  $\text{C}_{16}\text{TMA}$ , Tokyo Kasei Kogyo Co., Ltd.) and  $\text{H}_2\text{O}$  with a molar ratio of 1 : 1.1 : 92.8. A solution obtained by mixing of the raw materials was dried at 130 °C in air. XRD pattern of a sample as dried is shown in Figure 1a. Several peaks were observed at  $2\theta = 2.24^\circ$  ( $d = 43.8 \text{ \AA}$ ),  $4.16^\circ$  ( $23.6 \text{ \AA}$ ),  $6.06^\circ$  ( $16.2 \text{ \AA}$ ) and  $8.23^\circ$  ( $12.0 \text{ \AA}$ ), while no peaks at  $2\theta > 10^\circ$  were observed. This result indicates mesoscopic periodic structure of amorphous vanadium based materials formed along the template of surfactant micelles.



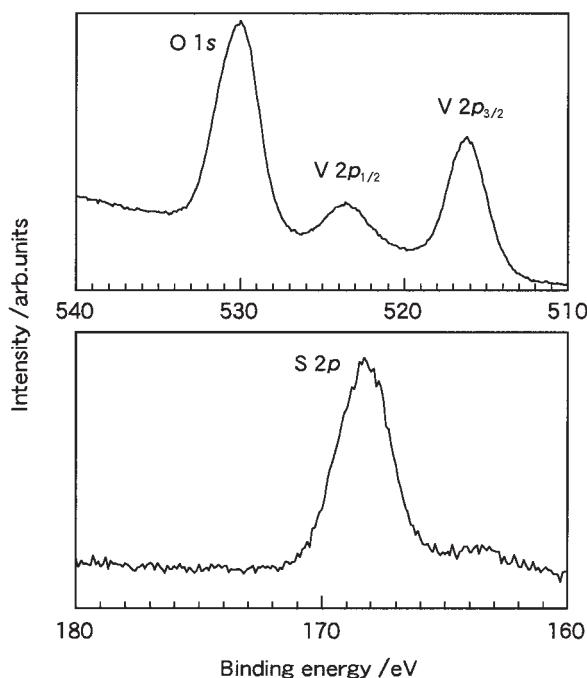
**Figure 1.** XRD patterns of a sample as dried with  $\text{C}_{16}\text{TMA}$  (a) and mesoporous  $\text{VOSO}_4$  based material heat-treated at 300 °C in flowing  $\text{N}_2$  (b).

Condition of thermal treatment to remove surfactant without degradation of periodic mesostructure in flowing  $\text{N}_2$  was examined using TG-DTA as shown in Figure 2. Weight loss accompanied by endothermic reaction around 70 °C, which was attributed to desorption of adsorbed water, was observed. Other weight losses accompanied by endothermic reaction around (1) 150–290 °C, (2) 320–390 °C and (3) 410–560 °C attributed to



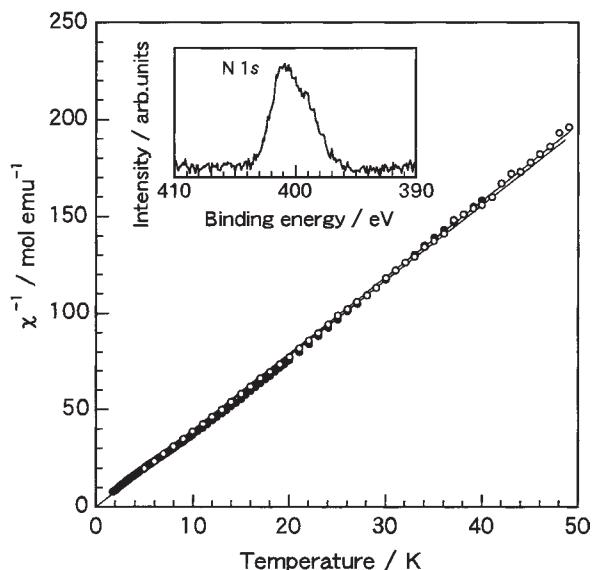
**Figure 2.** TG-DTA curves of a sample as dried with  $\text{C}_{16}\text{TMA}$  in flowing  $\text{N}_2$ .

thermal decomposition of surfactant were also observed. This figure indicates that thermal treatment above 560 °C is necessary to remove the surfactant completely. On the other hand, from the elimination of peaks around  $2\theta = 2^\circ$  in the XRD measurement, it was confirmed that the periodic mesostructure of the sample had already been damaged at 400 °C. Therefore, in this paper, characterization of a sample fired at 300 °C which still retained periodic mesostructure with XRD peaks at  $2\theta = 2.30^\circ$  ( $d = 38.4 \text{ \AA}$ ),  $7.81^\circ$  ( $d = 12.6 \text{ \AA}$ ) as shown in Figure 1b was carried out, in spite of a half surfactant remained in the sample. From XPS measurement, N 1s, C 1s and S 2p peaks besides V and O peaks were observed, but no Cl peaks were observed. Moreover, as shown in Figure 3, the binding energies of V  $2p_{3/2}$  and S 2p are in good agreement with that of VOSO<sub>4</sub> and SO<sub>4</sub><sup>2-</sup>, respectively.<sup>7</sup> Hence, the partially mesoporous vanadium based material fired at 300 °C consists of amorphous VOSO<sub>4</sub> network and residual organic component without vanadium oxychloride.



**Figure 3.** O 1s, V 2p and S 2p XPS spectra of mesoporous VOSO<sub>4</sub> based material heat-treated at 300 °C in flowing N<sub>2</sub>.

Figure 4 shows temperature dependence of reciprocal molar magnetic susceptibility measured by SQUID for the samples as-dried and fired at 300 °C. Number of moles of vanadium per 1 g of both samples were  $1.64 \times 10^{-3}$  mol and  $2.82 \times 10^{-3}$  mol, respectively, when a sample fired at 560 °C was assumed to be an anhydrous VOSO<sub>4</sub>. As shown in Figure 4, both samples obey Curie-Weiss law with  $\theta = 0$  K. This indicates that both the samples exhibit Langevin paramagnetism with free spins, which is different from  $\alpha$ - or  $\beta$ -VOSO<sub>4</sub> crystal. Effective magnetic moments  $\mu_{\text{eff}}$  of both samples were  $\mu_{\text{eff}} = 1.41 \mu_{\text{B}}$  and  $1.43 \mu_{\text{B}}$ , respectively. Since the values are close to theoretical spin-only value of V<sup>4+</sup> ( $3d^1$ ) ( $\mu_{\text{eff}} = 1.73 \mu_{\text{B}}$ ), oxidation number of



**Figure 4.** Reciprocal molar magnetic susceptibility vs temperature curves: ○ a sample as dried with C<sub>16</sub>TMA; ● mesoporous VOSO<sub>4</sub> based material heat-treated at 300 °C in flowing N<sub>2</sub>. Inset: N 1s XPS spectrum of the latter sample.

vanadium in both samples are considered to be V<sup>4+</sup> ( $3d^1$ ). However, deviation from the theoretical value is considerably larger than that of  $\alpha$ -VOSO<sub>4</sub> ( $\mu_{\text{eff}} = 1.69 \mu_{\text{B}}$ ) and  $\beta$ -VOSO<sub>4</sub> ( $\mu_{\text{eff}} = 1.54 \mu_{\text{B}}$ ) crystals with V<sup>4+</sup> ( $3d^1$ ).<sup>5,6</sup> The reasons are as follows; viz. a portion of oxygen of surface V-S-O network are thought to be substituted by nitrogen of surfactant due to large surface area attributed to mesostructure, which indicate existence of V<sup>5+</sup> ( $3d^0$ ). This assumption is based on the fact that broad peak which consists of the peaks more than two with different binding energies was observed in XPS spectrum of N 1s as shown in inset of Figure 4. In addition, this assumption may indicate that the elimination of magnetic transition in both the samples is attributed to hindrance of V-SO<sub>4</sub>-V superexchange interaction by V<sup>5+</sup>.

## References

- 1 C. T. Kresge, M. E. Leonowicz, W. J. Roth, J. C. Vartuli, and J. S. Beck, *Nature*, **359**, 710 (1992).
- 2 K. Yamada and S. Kohiki, *Physica E*, **4**, 228 (1999).
- 3 H. Higashijima, S. Kohiki, S. Takada, A. Shimizu, and K. Yamada, *Appl. Phys. Lett.*, **75**, 3189 (1999).
- 4 S. Kohiki, S. Takada, A. Shimizu, K. Yamada, H. Higashijima, and M. Mitome, *J. Appl. Phys.*, **87**, 474 (2000).
- 5 J. M. Longo and R. J. Arnott, *J. Solid State Chem.*, **1**, 394 (1970).
- 6 R. Kierkegaard and J. M. Longo, *Acta Chem. Scand.*, **19**, 1906 (1965).
- 7 C. D. Wagner, W. M. Riggs, L. E. Davis, J. F. Moulder, and G. E. Muilenberg, "Handbook of X-ray Photoelectron Spectroscopy," Perkin-Elmer Corporation, Minnesota (1979), p 56, p 70.